Temperature-dependent structural heterogeneity in calcium silicate liquids
Benmore, C. J.; Weber, J. K. R.; Wilding, M. C.; Du, J.; Parise, J. B.

Published in:
Physical Review B

DOI:
10.1103/PhysRevB.82.224202

Publication date:
2010

Citation for published version (APA):
Temperature-dependent structural heterogeneity in calcium silicate liquids

C. J. Benmore, 1,6 J. K. R. Weber, 1,2 M. C. Wilding, 3 J. Du, 4 and J. B. Parise 5

1X-ray Science Division, Advanced Photon Source, Argonne National Laboratory, Argonne, Illinois 60439, USA
2Materials Development, Inc., Arlington Heights, Illinois 60004, USA
3Institute of Mathematical and Physical Sciences, Aberystwyth University, Aberystwyth, Ceredigion SY23 3BZ, United Kingdom
4Department of Material Science and Engineering, University of North Texas, Denton, Texas 76203, USA
5Mineral Physics Institute and Department of Geosciences, Stony Brook University, Stony Brook, New York 11794-2100, USA

(Received 30 September 2010; published 7 December 2010)

X-ray diffraction measurements performed on aerodynamically levitated CaSiO3 droplets have been interpreted using a structurally heterogeneous liquid-state model. When cooled, the high-temperature liquid shows evidence of the polymerization of edge-shared Ca octahedra. Diffraction isosbestic points are used to characterize the polymerization process in the pair-distribution function. This behavior is linear in the high-temperature melt but exhibits rapid growth just above the glass transition temperature around 1.2Tg. The heterogeneous liquid interpretation is supported by molecular-dynamics simulations which show the CaSiO3 glass has more edge-shared polyhedra and fewer corner shared polyhedra than the liquid model.

DOI: 10.1103/PhysRevB.82.224202

PACS number(s): 61.20.-p, 61.05.cp, 64.70.P–

I. INTRODUCTION

Bulk liquids are often regarded as homogeneous materials partly due to their inherently disordered nature and also because many of their properties can be modeled by a uniform continuum. When liquids are supercooled, a dynamic heterogeneity occurs around the crossover from “normal” liquid to “complex” solidlike properties1 as the glass transition is approached. This coincides with a similar crossover from mode coupling theory at a temperature of T=1.2Tg between a primarily “ergodic” liquid to a mainly “nonergodic” glassy regime.2 Unlike the dynamic heterogeneity however, structural heterogeneity in a liquid at the atomic level is difficult to establish and to our knowledge there has been no definitive evidence for structural heterogeneity occurring in the pair-distribution functions of glass-forming liquids.2,3 This is despite the analysis of quenched glass structures that indicate atomic to nanometer scale structural heterogeneity must occur in the liquid state.4–7

Silica is classified as a strong liquid with an Arrhenius viscosity-temperature relation and only subtle differences occur between the liquid and glass structures.8,9 In contrast CaO-SiO2 liquids can be considered fragile based on viscosimetric growth of CaO6 octahedra in the high-temperature liquid to a mainly “ergodic” liquid to a mainly “nonergodic” glassy state.9,224202 Evidence of the polymerization of edge shared Ca octahedra. Diffraction isosbestic points are used to characterize the polymerization process in the pair-distribution function. This behavior is linear in the high-temperature melt but exhibits rapid growth just above the glass transition temperature around 1.2Tg. The heterogeneous liquid interpretation is supported by molecular-dynamics simulations which show the CaSiO3 glass has more edge-shared polyhedra and fewer corner shared polyhedra than the liquid model.

II. EXPERIMENTAL

Six compositions between 40 and 75 mol % CaO were investigated in the liquid state. The materials were made from calcium carbonate and silica powders that were mixed, calcined and consolidated into spheres in a laser hearth. Compositions containing less than 60 mol % CaO formed glasses when cooled. In this work we present only the results only for CaSiO3 in detail as the other glass-forming compositions all showed a similar behavior. The high-energy x-ray diffraction experiments on the calcium silicate liquids were performed using an aerodynamic levitator installed on beamline 11 ID-C at the Advanced Photon Source ( APS), USA.12 The temperature of the melts was controlled using a 240 W CO2 laser beam. The experiment was performed in transmission geometry with a 1 mm×1 mm x-ray beam of wavelength 0.1088 Å and diffraction patterns on isothermal melts were collected using a Mar345 image plate detector mounted orthogonal to the x-ray beam. Samples of 2–3 mm in diameter were made from binary (CaO)1−x(SiO2)x compositions containing x=40, 50, 60, 70, and 80 mol % CaO. The sample lost ~1% of its initial mass during the high-temperature levitation experiments in pure oxygen. Microprobe analysis on the recovered samples indicated that the composition change was <1%.

Calcium silicate liquids were studied at static temperatures ranging from ~800 to ~1900 °C in approximately 100 °C increments. The data were reduced using the software FIT2D (Ref. 13) and the x-ray structure factors, S(Q), obtained up to Q~19 Å−1 using the program PDFGETX2 (Ref. 14) see Fig. 1(a). The apparent temperature of the levitated sample was measured using an optical pyrometer that operated at a wavelength of 0.650 μm as described previously.12 The estimated errors associated with the temperature measurement arising from the emissivity correction and temperature gradients that occur in the high temperature melts are ±25 °C for T>1000 °C and ±50 °C at lower temperatures approaching the glass transition temperature.

Fast, transient cooling measurements were made on selected samples using a Perkin Elmer model 1621 x-ray area detector. This arrangement enabled a measurement of structure to be made in a period of 200 ms and a sequence of measurements were made as the liquid drop cooled from about 2000 to 600 °C. An optical pyrometer (Chino Corp-
FIG. 1. (Color online) Left: measured x-ray structure factors for CaSiO$_3$ over a temperature range that spans the superheated and supercooled liquid and the glass. Right: comparison of our CaSiO$_3$ data with previously measured x-ray structure factors for glassy and liquid CaSiO$_3$ and SiO$_2$ from the literature. (a) Shows liquid SiO$_2$ (scaled by 0.5) from Ref. 9 (magenta dotted line). (b) Our measured liquid structure factor for CaSiO$_3$ at 1650 °C (black solid line) is compared to that measured by Waseda and Toguri (Ref. 15) at 1600 °C (red dashed line). The solid black lines in (c) and (d) are the measured glass structure factors of CaSiO$_3$ at 25 °C (this study) compared to Taniguchi et al. (Ref. 16) [c] blue long dashed-dotted line] and Mead and Mountjoy (Ref. 17) [d] green short dashed-dotted line]. The curves are offset for clarity.

III. EXPERIMENTAL RESULTS

Our measured liquid and glass x-ray structure factors are shown in Fig. 1(a) and compared with curves digitized from the literature in Fig. 1(b). The difference between liquid CaSiO$_3$ at 1650 °C (this study) and 1600 °C from Ref. 15 is most likely due to a difference in composition, as noted previously. The preferential evaporation of CaO over a long (tens of minutes) period of time in the experiment described in Ref. 15 could leave a more silica-rich liquid than that of the CaSiO$_3$ starting material, as suggested by the similarity to pure liquid SiO$_2$ $S(Q)$ as illustrated in Fig. 1(b). The shape of our glass x-ray structure factors are similar to those measured by Taniguchi et al. except in the low-$Q$ region [Fig. 1(c)] but are in good agreement with the more recent data of Mead and Mountjoy [Fig. 1(d)].

The real space pair-distribution functions $D(r) = 4\pi\rho[\rho(r)-1]$ with the bulk density removed were obtained for liquid CaSiO$_3$ at 1900 °C and 1033 °C and are shown in Fig. 2. Previous x-ray measurements on the glass, have established the presence of SiO$_4$ tetrahedra and the CaO peak at $\sim 2.43 \pm 0.02$ Å. The structure of the glass at a composition close to CaSiO$_3$ (with the addition of 3% Al$_2$O$_3$) has been studied in exceptional detail using a double difference neutron isotopic substitution method. Gaskell et al. provided evidence that the distribution of calcium atoms was not random in the glass but exhibited a well-defined Ca-Ca peak at 3.8 Å bearing many characteristics of the crystalline form. The structures of crystalline pyroxenes such as wollastonite can be considered as a close packed oxygen sublattice with Ca occupying octahedral interstices in alternate layers in a fcc model. This structure has distinguishable domains comprising of sheets of edge shared CaO$_6$ octahedra which are parallel and interleaved with linked SiO$_4$ tetrahedra. Although the recent model by Mead and Mountjoy and our MD simulations show that the Ca polymerization in the glass need not be quasicrystalline in nature and branched winding chains of CaO polyhedra are also consistent with the diffraction data.

IV. MOLECULAR-DYNAMICS SIMULATIONS

The contributions of the six pair distribution functions to the measured CaSiO$_3$ x-ray functions were established using classical MD simulations using potentials based on the point charge Born model. Partial covalency of the Si-O bonds was modeled through the partial charge on the cations and anions. The short range interaction consists of an exponential repulsion term and a dispersion attraction term, in the form of Buckingham

$$V_{ij}(r) = A_{ij} \exp(-r_{ij}/\rho_{ij}) - C_{ij}/r_{ij}^6,$$

where $r_{ij}$ is the distance between ion $i$ and $j$. $A$, $\rho$, and $C$ are parameters. The potential parameters were obtained by com-
TABLE I. Atomic charges and Buckingham potential parameters.

<table>
<thead>
<tr>
<th>Pairs</th>
<th>A (eV)</th>
<th>ρ (Å)</th>
<th>C (eV Å²)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Si²⁺-O⁻¹·²</td>
<td>13702.905</td>
<td>0.193817</td>
<td>54.681</td>
</tr>
<tr>
<td>O⁻¹·²-O⁻¹·²</td>
<td>2029.2204</td>
<td>0.343645</td>
<td>192.58</td>
</tr>
<tr>
<td>Ca¹⁺²-O⁻¹·²</td>
<td>7747.1834</td>
<td>0.252623</td>
<td>93.109</td>
</tr>
</tbody>
</table>

Incorporating ab initio calculations with the empirical fitting of related mineral structure and physical property data. This method has been successfully applied to a number of alkali silicate, rare-earth aluminate, and silicate simulations. The partial atomic charges and Buckingham potential parameters are shown in Table I.

At short interatomic distances, the Buckingham potential is known to have an unphysical deep potential well that can lead to instabilities of the simulation system. This is caused by the over-riding of the power term to the exponential term at short distances. This was corrected by splicing a power term while maintaining the potential, force, and derivative of force to be continuous at the splicing point. The splicing point was chosen to be where the third derivative of the potential energy was zero. This modification corrects the unphysical potential well at short distance but leaves the original potential unchanged at longer distances. The correction is critical in the simulation of the molten state since the high kinetic energy of ions in the melt can overcome the barrier and reach the deep potential well.

The MD simulations were performed using the constant mass (N, the number of atoms), pressure P and temperature T (NPT) ensemble with 1 fs time steps for 80,000 steps. Each cubic simulation cell contained 2500 atoms and with the initial simulation cell size determined based on the experimental density of the glass. At each of the eight simulation temperatures, using the constant mass, volume V and total energy (NVE) ensemble run of 80,000 steps followed the NPT simulations. Structure analyses were obtained by averaging 400 structure snapshots of the final 20,000 steps under NVE. For our potential the melting temperature of the model system was 4000 K so structure factors were extracted at $\Delta T_m$, $0.875\Delta T_m$, $0.75\Delta T_m$, and $0.625\Delta T_m$ to provide the same effective temperature range as the supercooled liquid experimental data. The simulations confirm that the $2.34 \text{ Å}$ peak in the x-ray pattern arises mainly from the Ca-O peak and shows a high-$r$ contribution from the O-O correlations. Whereas the main contributions to the $3.50 \text{ Å}$ peak comes mainly from the Ca-Si and Ca-Ca partial pair-distribution functions.

V. DISCUSSION

A. High-temperature liquid state

The concept of diffraction isosbestic points was introduced by Johnson, to explain the linear dependence of the differential distribution function with concentration in Si-rich $\text{Si}_4\text{Se}_8(1-x)$ glasses. Isosbestic points have traditionally been associated with identifying a two-state liquid system using optical spectroscopy. Diffraction isosbestic points are common points in the diffraction pattern and Tse et al. have extended Johnson’s ideas to argue that a system exhibiting these points as a function of temperature can be interpreted in terms of a two-state model. We note that since all of our data were Fourier transformed at the identical value of $Q = 19 \text{ Å}^{-1}$ with a Lorch modification function this does not enter into the analysis. Furthermore, although diffraction isosbestic points for $T > 1000 \text{ °C}$ are observed directly in the differential distribution functions, $D(r)$, shown in Fig. 2, they become much more apparent in the difference function $\Delta D(r) = D_T(r) - D_0(r)$ shown in Fig. 3, where the highest temperature $D_0(r)$ at 1900 °C has been subtracted. Diffraction isosbestic points are observed for liquid $\text{CaSiO}_3 (T > 1000 \text{ °C})$ at 1.48, 1.63, 2.15, 2.52, 3.13, 4.47, 5.80, and 7.28 Å. For calcium silicate liquids we argue that Johnson’s clustered unit model applies in this temperature regime in the form of aggregation of Ca-rich atom regions (by analogy with the compositional dependence in Ref. 28). Based on the peaks in the liquid diffraction data for $\text{CaSiO}_3$ this implies the existence of two distinct domains on a length scale of ~3–18 Å: one domain comprising of edge shared CaO$_6$ octahedra and the other a mixture of CaO polyhedra and SiO$_4$...
tetrahedra, these domains are presumably corner shared by oxygen atoms at the interface. For temperatures \( T_g < T(\degree C) < 1000 \) the diffraction isosbestic points are no longer observed, consistent with the transition from normal to complex behavior around \( T = 1.2 T_g \). The onset of solid like behavior is associated with rapid peak growth associated with Ca polymerization with decreasing temperature. It is important to note that the existence of isosbestic points do not definitively prove the system is two phase; the sharpening of intramolecular correlations in a molecular liquid upon cooling for example, could also produce the same effect due to narrowing zero point motions. However, strong arguments for the heterogeneous clustered unit model can be made due to the observation of the increased number of edge shared CaO octahedra upon cooling liquid CaSiO\(_3\) and because of the supporting evidence from MD simulations (see Fig. 3).

The MD model shows that at the melting temperature a few threefold and fivefold coordinated Si exist but 98.5\% of these polyhedra remain corner shared and only 1.5\% are edge shared. By 0.75\( T_m \) all the SiO\(_4\) units are corner shared and in the room-temperature glass all Si ions are tetrahedrally coordinated by oxygen. The MD model shows that the number of edge and face shared Ca polyhedra increase steadily with decreasing temperature indicating Ca polymerization occurs upon supercooling the melt, see Fig. 4.

**B. Below 1.2\( T_g \)**

The main local structural changes that occur upon cooling the melt are at 2.34 and 3.50 Å in the x-ray differential distribution function (Fig. 3) and we interpret the growth of these peaks upon cooling as the increase in size of edge-shared octahedral clusters or chains. Integrating these two peaks in the \( \Delta D(r) \) functions between associated isosbestic points we can obtain an estimate of the rate of growth of the CaO\(_6\) and Ca-Ca edge-sharing octahedral units in the melt as a function of temperature, see Fig. 4. The same integration limits were also used below 1000 °C for consistency although the isosbestic points are no longer present introducing a small error. Given that, for both peaks the integrals can be fitted using a linear term at high temperatures and a non-linear term at low temperatures. This analysis implies that there is a linear growth of chains and/or (possibly) sheets of CaO\(_6\) in the liquid between 1900 and 1000 °C but the aggregation increases more rapidly as the glass transition temperature is approached and is concomitant with a steep rise in viscosity (Fig. 5). Similar curves were obtained for the data for all the glass forming compositions studied but only linear behavior is observed for the high CaO content samples which could not be vitrified.

**C. Fast-cooling experiments**

In order to probe the low-temperature region a separate set of experiments were performed using the area detector. The CaSiO\(_3\) melts were quenched by switching the laser off and sequential measurements were taken every 200 ms. The results shown in Fig. 6 around 1000 °C are therefore intriguing since they suggest the small deviation in structure between the isothermal and fast quenched CaSiO\(_3\) samples which begins at ~1100 °C is real.

The fast cooling measurements also show a distinct plateau as \( T_g \) is approached that is absent in the slow-cooling measurements. This difference is attributed to the higher fictive temperature expected in the quenched sample compared to one that is vitrified by slow cooling, i.e., different thermal histories. The differences observed in the present work also provide a possible explanation of the differences in structure measured in prior work by Taniguchi et al.\(^{16}\) and Mead and Moutinjov.\(^{17}\) Further fast-cooling structural studies are planned on other systems to see if these features are ubiquitous in quenched glass-forming liquid oxides.

**VI. SUMMARY**

Static and fast cooling high-energy synchrotron x-ray structure factor measurements have been performed on aero-

---

**FIG. 4.** (Color online) Changes in Ca-polyhedral connectivity of the liquid/glass MD model of CaSiO\(_3\) as a function of temperature.

**FIG. 5.** (Color online) The integrals of the differences between the static state differential distribution functions, \( \Delta D(r) \) measurements (see text), shown in Fig. 3 (top) for liquid CaSiO\(_3\). The integrals were taken over the region between the isosbestic points for the 2.34 and 3.50 Å peaks in \( \Delta D(r) \). The symbols represent different experimental runs. Both curves are fitted with a linear (blue dashed) line which has been extended to lower temperatures by a dashed blue line (see Fig. 6 for more extensive measurements in this region). The glass measured at 25 °C was taken as a representative liquid structure at \( T_g = 781 °C \) for CaSiO\(_3\) (Ref. 31) (solid symbols). The 1.2\( T_g \) line is calculated in Kelvin but shown in degrees Celsius. Inset: viscosity-temperature relation of liquid CaSiO\(_3\) (blue squares) (Refs. 10 and 11) and SiO\(_2\) (circles) (Ref. 32).
The integrals were taken over the region between the isosbestic points for the 3.50 Å Ca-Ca peak in ΔD(r). The fast cooling measurements (solid red squares) are compared to isothermal measurements (open symbols from Fig. 5). The dotted line represents Tg (Ref. 31) and the dashed line 1.2Tg (992 °C). The 1.2Tg line is calculated in Kelvin but shown in degrees Celsius for consistency.

FIG. 6. (Color online) The integrals of the differences between the differential distribution functions, ΔD(r), for liquid CaSiO3.

The integrals were compared to isothermal measurements to illustrate the isosbestic points for the 3.50 Å Ca-Ca peak in the ΔD(r) function. The fast cooling measurements (solid red squares) are compared to isothermal measurements (open symbols from Fig. 5). The dotted line represents Tg (Ref. 31) and the dashed line 1.2Tg (992 °C). The 1.2Tg line is calculated in Kelvin but shown in degrees Celsius for consistency.

ACKNOWLEDGMENTS

We appreciate the efforts of W. Woerner in performing the probe analyses, and for access to infrastructure provided by H. Nekvasil and D. H. Lindsley at Stony Brook used for preparing the precursor materials for this study. This work was supported by the U.S. DOE, at Argonne National Laboratory under Contract No. DE-AC02-06CH11357, by NSF under Grant No. DMR-0800415, and NSF under Grants No. DMR-0907593 and No. DE-SC0002510. J.B.P. is grateful to the UK Fulbright Program for support during a sabbatical stay at Edinburgh University.

*Corresponding author.

(2005).